Influence of phosphate anion adsorption on the kinetics of oxygen electroreduction on low index Pt(hkl) single crystals†

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The detrimental effects of phosphate anion adsorption on the oxygen reduction reactions (ORR) on low index Pt single crystal electrodes were studied in 0.1 M perchloric acid by using a hanging meniscus rotating disk electrode in the presence of varied concentrations of H₃PO₄. The kinetic current for ORR decreased dramatically on Pt(100), Pt(110), Pt(111), and PtSn(111) even with the addition of a small amount (1 mM) of H₃PO₄ into the perchloric acid solution, most probably due to the adsorption of phosphate anions onto the Pt active sites that impeded the electroreduction of O₂. Remarkably, the extent of decline was found to vary with the specific single crystal surface, following the order of Pt(111) > PtSn(111) > Pt(110) ~ Pt(100). Consistent behaviors were also observed in Tafel analysis and in electrochemical impedance spectroscopic measurements. Within the present experimental context, Pt(110) was found to be the optimal crystal surface for ORR in phosphoric acid fuel cells with the smallest charge transfer resistance, whereas the poisoning effects of phosphate anion adsorption were the most pronounced on Pt(111), most likely because the phosphate anions primarily adsorbed on the 3-fold sites on the Pt(111) faces, as manifested in in situ X-ray absorption spectroscopic measurements.

1. Introduction

Phosphoric acid fuel cells (PAFCs) have been commercialized successfully and used for stationary applications with a combined heat and power efficiency of about 80%. However, continuous improvements are still imperative if PAFCs are to hold their positions in the market or penetrate into other areas with moving targets in terms of cost and efficiency. Currently, the most favored materials for PAFC cathode catalysts are Pt in high-surface-area forms (i.e., nanosized particles) supported on carbon. Over the years, extensive studies have been carried out for the preparation of ultrasmall Pt particles so as to maximize the surface to volume ratio. Recent improvements in membrane electrode assembly (MEA) technologies have made it possible to achieve ready accessibility to the electrolyte and gaseous phases for all catalyst-containing parts, and a three-phase contact between the electrolyte film, gaseous reactant, and the electronically conducting catalyst support.

However, one major factor that limits the performance of PAFCs is the sluggish kinetics of oxygen reduction reactions (ORR) in H₃PO₄, and this is attributed primarily to the poisoning effects of phosphate ion adsorption (H₂PO₄⁻) on ORR activity. As the adsorption of anions is structure-sensitive, the ORR performance is anticipated to vary sensitively with the surface structure of the catalysts. Furthermore, the adsorption may be reversible (e.g., adsorption of hydroxyl species from water activation) or partially reversible (e.g., adsorption of sulfate, phosphate, halogen, etc). The effect of such adsorption on the pathway of ORR, i.e., site blocking or rendering alternative lower energy transition states to be formed, is of great interest in designing the next-generation electrocatalysts with improved performance. At a fundamental level, understanding electrocatalytic pathways on single crystal surfaces provides important insights. However translating these results to experimental observations on highly dispersed nanocluster polycrystalline materials is somewhat difficult especially in the size range below 2 nm, where there is an exponential increase in non-bulk planes such as edge, kink and vertex sites. The wealth of literature however points to improved electrocatalysis for ORR for particle sizes significantly higher than 4-5 nm where bulk planes (100, 111, 110, etc.) predominate.

Adsorption of anions such as (bi)sulfate, chloride, and hydroxyl on Pt single crystals and Pt nanoparticles in an electrochemical environment has been extensively studied. However, there are only a few reports about phosphate anion adsorption on Pt, possibly because of the polyprotic nature of phosphoric acid that makes the study complicated. Weber et al. claimed that the adsorption of phosphate species was...
insensitive to the Pt surface geometry with mainly H$_2$PO$_4^-$ adsorbed in the low potential region and a mix of H$_2$PO$_4^-$ and HPO$_4^{2-}$ adsorbed in the high potential regime. In contrast, Tanaka et al.\textsuperscript{19} believed that the phosphate anion adsorption was structure-dependent as manifested by the variation of ORR in H$_2$PO$_4$ solutions with electrode structure. It should be noted that there exists considerable differences in the published literature on the exact nature and mode of phosphate adsorption on Pt surfaces. For instance, it has been found that both H$_2$PO$_4^-$ and HPO$_4^{2-}$ can be adsorbed on the Pt(111) and Pt(100) surfaces. In one study based on \textit{in situ} FTIR measurements, it was argued that the stretching vibrations of the H$_2$PO$_4$ ion occurred through two oxygen atoms with C$_{2v}$ symmetry and those of HPO$_4^{2-}$ ion through three equivalent oxygen atoms with C$_{3v}$ symmetry.\textsuperscript{18} Yet in another study, Tanaka et al. proposed that the adsorption of phosphate anions occurred \textit{via} three oxygen atoms on the Pt(111) surface in the three-fold sites and only one or two oxygen atoms were involved in the adsorption on Pt(100) and Pt(110) surfaces.\textsuperscript{19} In the context of these prior reports it is important to note that in the concentration range of 1 to 100 mM the predominant species is expected to be dihydrogen phosphate anion (H$_2$PO$_4$) (concentration six orders of magnitude higher than that of HPO$_4^{2-}$) when considering the corresponding dissociation constants (5 orders of magnitude difference) at room temperature at pH 0 to 1.

A few studies have been reported in the literature about the kinetics of ORR on single crystal Pt surfaces in H$_2$PO$_4$. Tanaka et al.\textsuperscript{19} demonstrated a four-electron process for ORR with first-order kinetics on three low-index Pt single crystals in 0.1 M H$_2$PO$_4$ and concentrated 85% H$_2$PO$_4$. An order of Pt(110) > Pt(100) > Pt(111) in terms of catalytic activity towards ORR was provided in their study. Kadiri et al.\textsuperscript{8} found that the ORR activity varied on different Pt single crystals with the sequence of Pt(110) > Pt(100) > Pt(111) in 1 M H$_2$PO$_4$. They believed that ORR was orientation and structure-sensitive in H$_2$PO$_4$ solution and anion adsorption might change according to the surface structure of the single crystal electrodes. However, insufficient proofs have been given in the above literature to draw the conclusion that the order of catalytic activity noticed is purely caused by structure-dependent phosphate anions adsorption given the fact that ORR is a structural dependence reaction itself on Pt surface.\textsuperscript{12,13,20–22}

Strategically, deterioration of ORR kinetic current ($\Delta I_k$) can be assessed by conducting ORR measurements in the presence and absence of H$_2$PO$_4$ on a Pt single crystal with unique surface orientation. Comparison of $\Delta I_k$ on different Pt single crystals will provide us with the information on how the specific adsorption interferes with ORR on Pt with various surface morphologies, where only the structure-dependent adsorption of the phosphate anion plays the contributing role.

Within this context, the goal of this work is to further our understanding of the effects of phosphate adsorption on oxygen reduction by using single crystal Pt electrodes in 0.1 M HClO$_4$ with the addition of different amounts of H$_2$PO$_4$ (1 to 100 mM). Incremental dosing of H$_2$PO$_4$ was (a) to detect any changes in anion adsorption mode and (b) to determine the threshold concentration beyond which no further poisoning effect is observed. A hanging meniscus rotating disk (HMRD) electrode was used to quantitatively determine the kinetic parameters of ORR on three low-index Pt single crystals (Pt(100), Pt(110), and Pt(111)) as well as a surface modified single crystal (PtSn(111)) electrode. The poisoning effects of phosphate adsorption on ORR charge transfer dynamics were then examined by voltammetry and electrochemical impedance spectroscopy (EIS). The choice of well ordered PtSn(111) was aimed at elucidating the effect of (a) alloying from the perspective of a crystal plane with the highest atomic density (111), and (b) Sn being most easily oxidized, its presence in an ordered atomic array in the context of its effect on phosphate anion adsorption (concomitant comparison with Pt(111)). These investigations were aimed at providing pointers in better designs of polycrystalline nanoparticles whose relative population of bulk crystal planes and nature of available sites can be engineered \textit{via} adjustment of particle size and choice of alloying elements. \textit{In situ} X-ray absorption spectroscopy was employed to elucidate the site-specific adsorption of phosphate anions on Pt single crystal surfaces.\textsuperscript{14,23–26}

2. Experimental section

2.1 Preparation of single crystals

The preparation and cleaning of the Pt single crystals were performed in a stainless steel ultra-high vacuum (UHV) chamber with a base pressure of 1.0 × 10$^{-10}$ Torr. This system is equipped with low energy ion scattering (LEIS), high-resolution X-ray photoelectron spectroscopy (XPS), and low energy electron diffraction (LEED). The crystals were cleaned by cycles of Ar$^+$ sputtering (2.0 × 10$^{-5}$ Torr, 500 eV) and annealed at 900 K in 5.0 × 10$^{-8}$ Torr of oxygen. Flushing to 1100 K was required to remove residual oxygen and sharp LEED patterns were obtained to ensure good quality of the single crystal surfaces. The electrodes were typically cooled down in a H$_2$/Ar atmosphere to lift the reconstruction when the electrodes were still hot. A tin doser was also used to evaporate Sn on Pt(111) to generate a Sn/Pt(111, 2 × 2) alloy surface, followed by annealing to different temperatures as described by Xu \textit{et al.}\textsuperscript{25} LEIS and XPS were also performed to ensure that all surfaces were free of even trace levels of impurities. The cleaned crystals were then transferred from the UHV chamber to an argon gas-filled box. In this box, a working electrode was made by mounting a crystal onto the top of a rotating disc electrode through a spot-welding method. Briefly, a small metal wire was welded onto the low edge of the crystal and side of the rotating disc electrode. This epoxy-free working electrode was very stable during rotation. A hanging meniscus rotating disc technique was used in the electrochemical measurements. The height of the meniscus was carefully controlled to prevent the side of crystals from contacting the solution (lateral wetting).

Because of irreversible adsorption of phosphoric acid on Pt single crystal surfaces which might lead to adsorption-induced surface reconstruction, the Pt surfaces were prepared independently for each electrochemical measurement. Surface cleanliness and LEED patterns were checked before each
experiment. Typically, each electrode was treated three times before being used in electrochemical measurements. The surface areas of these single crystal electrodes were Pt(100), 0.810 cm²; Pt(110), 0.712 cm²; Pt(111), 0.568 cm²; and PtSn(111), 0.502 cm².

2.2 Electrochemical measurements

Electrochemical measurements were conducted in a standard three-compartment electrochemical cell at room temperature using a rotating disk electrode (RDE) setup from Pine Instruments connected to an Autolab potentiostat (Ecochemie Inc., Model-PGSTAT 30). All potentials were measured with respect to a sealed hydrogen reference electrode (RHE) made from the same electrolyte used in the ORR experiments. A platinum wire was used as the counter electrode. The electrolytes were prepared with HClO₄ (GFS Chemicals) and purified H₃PO₄ (Sigma-Aldrich) using standard procedures described in the literature.

Voltammetric measurements were carried out with Pt(111), Pt(110), Pt(100), and PtSn(111) electrodes in either deoxygenated or oxygen-saturated 0.1 M HClO₄ with the addition of different amounts of H₃PO₄ (1 to 100 mM). EIS measurements were conducted at +0.85 V (vs. RHE) by using an EG&G PARC Potentiostat/Galvanostat (Model 283) and Frequency Response Detector (Model 1025). The impedance spectra were recorded between 100 kHz and 10 mHz with the amplitude of the ac signal at 5 mV.

3. Results and discussion

3.1 Cyclic voltammetry

Fig. 1 shows the cyclic voltammograms of (a) Pt(100), (b) Pt(110), (c) Pt(111) and (d) PtSn(111) single crystal electrodes in 0.1 M HClO₄ with the addition of different amounts of H₃PO₄ (note that at this pH, ~1, the dominant ionization product of H₃PO₄ is H₂PO₄⁻). It can be seen that before the addition of H₃PO₄ all three electrodes exhibited voltammetric features that were consistent with those reported in the literature, indicating well-defined crystalline
surfaces (further confirmed by the corresponding LEED measurements, not shown). Additionally, symmetrical voltammetric peaks appeared in the hydrogen UPD ($H_{\text{upd}}$) region of $E < +0.4 \text{ V}$. 

Upon the addition of $H_3PO_4$, the (a) Pt(100) and (b) Pt(110) electrodes both exhibited a diminishment of the double layer currents, and the decrease was more pronounced at Pt(100) than at Pt(110). To account for these observations, it should be noted that the onset potential for phosphate anion adsorption on the Pt(100) surface was no less than +0.3 \text{ V (vs. RHE)},^{18} Thus, the peaks seen in the $H_{\text{upd}}$ region in panel (a) might be attributed to H adsorption on the (100) bi-dimensional sites, and the new anodic peak at ca. +0.4 V most likely arose from surface reconstruction induced by the adsorption of phosphate ions.^{30} A similar observation was reported with Au(100) that was found to undergo surface reconstruction (1 × 1 ↔ hexagonal) after anion adsorption,^{18,30} leading to the appearance of extra positive charges on the electrode surface due to a change of the potential of zero charge (pzc). In contrast, no surface reconstruction was observed at Pt(110) (panel b). Yet, the suppression of the $H_{\text{upd}}$ peak at ca. +0.23 V and the growth of the one at +0.14 V might be ascribed to the stronger adsorption of phosphate anion at more positive electrode potentials.

The behaviors of the Pt(111) electrode were somewhat different. From panel (c), it can be seen that the voltammetric currents at Pt(111) were virtually invariant in the $H_{\text{upd}}$ region with the addition of $H_3PO_4$, where the anomalous state or butterfly envelope feature^{18} remained very well defined and shifted cathodically with increasing concentrations of $H_3PO_4$. However, there was a pair of new peaks emerging at ca. +0.55 V which was attributable to the specific adsorption/desorption of phosphate ions.^{19,31} Similar findings were reported in Nikolic’s work in the determination of the valence state of phosphate anions adsorbed on Pt(111).^{31} Notably, an appreciable change was seen even after the addition of 1 mM $H_3PO_4$ into the electrolyte solution. Further addition of $H_3PO_4$ caused only slight changes of the voltammetric currents and adsorption equilibrium was reached after the addition of 50 mM $H_3PO_4$. The adsorption of phosphate anions occurred after the desorption of H and ceased at the onset potential of OH adsorption. Notably, the inhibition to water activation^{12} and Pt oxide formation^{12,33} (positive shift of onset potential) on Pt(111) due to the adsorption of phosphate anions appears to be more prominent than that on Pt(100) and Pt(110), which will be discussed in more detail in the next section of oxygen reduction.

The differences observed in the voltammetric responses for the different bulk crystal planes mentioned above in the presence of phosphate anions might be further understood by considering the different surface structures of the Pt single crystals. As an FCC crystal, Pt exposes a closed packed hexagon in the (111) face, where adsorption might occur at the atop, bridge, and hollow sites, whereas atoms on the (110) face exhibit an up and down arrangement, and the ideal (110) face is not energetically favorable, which may be reconstructed to form a (2 × 1) structure with missing rows in the (110) direction. This reconstructed surface actually exposes two kinds of (111) planes, which are perpendicular to each other with only two rows of atoms exposed on the (111) surface. The space hindering effect prevents phosphate anions from adsorbing on these two (111) planes simultaneously. Atoms in the (100) surface, however, display a square arrangement, which is not energetically favorable either. Surface reconstruction of the ideal (100) surface forms a quasi-hexagonal packing structure,^{34–36} which might be removed by the adsorption of small molecules such as NO, CO, etc.^{27}

In order to investigate phosphate anion adsorption on alloyed Pt single crystals, we also prepared a PtSn(111, 2 × 2) surface alloy by evaporating Sn onto Pt(111) in UHV, followed by thermal annealing at 1000 K to obtain a clear and sharp (2 × 2) LEED pattern. The voltammetric features of PtSn(111) in 0.1 M HClO$_4$ and 0.1 M HClO$_4$ + 100 mM $H_3PO_4$ (Fig. 1(d)) were very similar to those depicted in Fig. 1(c) for Pt(111).

In summary, based on the voltammetric studies, it can be seen that the adsorption of phosphate anions depends highly on the facets and orientation of the Pt single crystal electrodes, and the adsorption occurs most saliently on Pt(111), as manifested in in situ X-ray absorption spectroscopy measurements (Figs. S1 and S2, ESI†), which is anticipated to lead to severe inhibition of water activation and oxide species formation, as discussed in further detail below.

### 3.2 ORR study

To minimize the complication arising from the polyprotic nature of $H_3PO_4$,^{17} we opted to maintain a constant solution pH while keeping the concentration and diffusion coefficient of $O_2$ constant for ORR measurements.^{19} Fig. 2 shows the RDE voltammograms of ORR at Pt(100), Pt(110), Pt(111), and PtSn(111) single crystal electrodes in 0.1 M HClO$_4$ (pH ≈ 1) at 900 rpm. The ORR curves on these electrodes at other rotation rates can be seen in Fig. S3 (ESI†). Three regions may be identified, namely, the kinetics controlled region ($> +0.9 \text{ V}$), mixed diffusion-kinetic limitation region ($+0.7 \text{ V to } +0.9 \text{ V}$) and a diffusion-controlled region.

![Fig. 2 RDE voltammograms for the Pt(100), Pt(110), Pt(111), and PtSn(111) electrodes in $O_2$ saturated 0.1 M HClO$_4$ at 900 rpm. The current density was calculated by normalizing the voltammetric current to the geometrical area of the electrode. DC ramp 20 mV s$^{-1}$](image-url)
(< +0.7 V). However, the diffusion-controlled region on Pt(100) is not well defined, especially at high rotation rates, which may be accounted for by (i) the surface tension at the Pt(100) surface that renders it difficult for O₂ to transport from the solution to the electrode surface through the thin film of electrolyte, and (ii) traces of Cl⁻ and NO₃⁻ in the solution that are prone to adsorb on Pt(100) and hence affect ORR. In addition, the small decrease of limiting currents at \( E < +0.3 \text{ V} \) with (a) Pt(100) and (c) Pt(111) might be attributable to H adsorption on the Pt surface active sites. Such a decrease did not happen on (b) Pt(110). Additionally, of note is that the shape of the current–potential curves on (c) Pt(111) and (d) PtSn(111) is very similar, again, signifying a Pt-rich surface on the PtSn(111) electrode.

The ORR current density \( j \) can be expressed by the Koutecky–Levich equation,

\[
\frac{1}{j} = \frac{1}{j_k} + \frac{1}{j_L}
\]

(1)

where \( j_k \) is the kinetic current density, \( j_L \) is the diffusion limiting current density. Due to the hydrodynamic behaviors of the hanging meniscus rotating disk system, the diffusion limiting current density \( (j_L) \) to be described by using a modified Levich equation,

\[
j_d = 0.620nFD^2C_or^2/\omega[1 - 2KR_0^{-1}(\nu/\omega)^2]
\]

(2)

where \( n \) is the number of electron transfer, \( \omega \) is the rotation rate, \( F \) is the Faraday constant, \( D \) is the diffusion coefficient of O₂ in 0.1 M HClO₄ solution, \( \nu \) is the kinematic viscosity, \( C_O \) is the concentration of molecular oxygen, \( R_0 \) is the geometric radius of the electrode, and \( K \) is a constant related to the meniscus height.

The plots of \( j_k \) vs. \( \omega^{1/2} \) are depicted in Fig. 3 based on data collected from Fig. S3 (ESI†), which show a linear plot and negative intercept for all electrodes, as anticipated from eqn (2). The experimental slopes are similar to the theoretical Levich plot with \( n = 4 \) (Table 1), indicating that the ORR followed the 4-electron reduction pathway to water on all the Pt single crystal electrodes. The negative intercepts indicate that the lateral wetting effect was minimal.

Further analysis based on eqn (1) and (2) shows that the plots of \( j^{-1} \) vs. \( j_L^{-1} \) are all linear and parallel at different potentials (Fig. 4), demonstrating that the kinetics of ORR on these single crystal electrodes is first order with O₂. The intercept of each plot reflects the value of \( j_k \) at a specific electrode potential, which may be correlated with the reaction rate constant \( (k_E) \) by eqn (4).

\[
j_k = nFk_EC_O
\]

(3)

The \( k_E \) values of the varied single crystal electrodes in different acid systems are summarized in Table 1, where one can see that on all these Pt single crystals, \( k_E \) decreases with increasing \( H_3PO_4 \) concentration, indicating that the ORR reaction becomes more and more sluggish due to the adsorption of phosphate anions.

The effects of phosphate anions adsorption on ORR on different crystallographic planes are further manifested in Fig. 5 and 6. Both anodic and cathodic ORR currents on Pt(111) in 0.1 M HClO₄ and 0.1 M HClO₄ + 50 mM \( H_3PO_4 \) are shown in Fig. 5 (those for Pt(100) and Pt(110) were included in Fig. S4 (ESI†)). Two conclusions can be drawn from these data: (a) in the presence of \( H_3PO_4 \) in the electrolyte solution, substantial overpotentials appeared for ORR. (b) It has been well-known that the hysteresis between cathodic scan and anodic scan on ORR curves on Pt electrodes is due to the formation of platinum oxide. As a result, phosphates do not compete with oxides during adsorption since only a slight change is observed in hysteresis in the presence of phosphate anions. Fig. 6 depicts the RDE voltammograms at the four electrodes in oxygen-saturated 0.1 M HClO₄ with the addition of various amounts of \( H_3PO_4 \). It can be seen that in all cases, the half-wave potentials \( (E_1/2) \) exhibited an apparent cathodic shift with increasing concentration of \( H_3PO_4 \), and the shift varied with the specific crystalline structure of the electrode. For instance, at 100 mM \( H_3PO_4 \), the shift of \( E_1/2 \) is 141 mV at Pt(111) (panel c), 99 mV at Pt(100) (panel a), 95 mV at Pt(110) (panel b), and 102 mV at PtSn(111) (panel d). That is, the poisoning effects of phosphate adsorption on the electrocatalytic reduction of oxygen decrease in the order of Pt(111) > PtSn(111) > Pt(100) > Pt(110). This order is also manifested in Fig. 7 which compares the kinetic current density \( (j_k) \) at +0.85 V of these four electrodes in the absence and presence of 100 mM \( H_3PO_4 \). It can be seen that in all cases, the addition of \( H_3PO_4 \) led to substantial diminishment of the kinetic current density. For instance, at Pt(100) electrode, the kinetic current decreased by about 72%; at Pt(100), 76%; at Pt(111), 88%; and at PtSn(111), 83%.

Consistent behaviors were observed in Tafel plots. Fig. 8 depicts the mass-transport corrected Tafel plots of the four single crystal electrodes in 0.1 M HClO₄ and 0.1 M HClO₄ + 100 mM \( H_3PO_4 \), where two slopes may be identified at low and high current density (Table 1). A similar response was also observed for ORR on single crystal and polycrystalline Pt electrodes. The difference in Tafel slopes observed in
0.1 M HClO$_4$ can be used to quantify the effect of oxide formation without the interference of anions in the electrolyte since perchloric acid is believed to show minimal poisoning to Pt electrodes.$^{8,12,22}$ As a consequence, two typical Tafel slopes of 60 mV per decade and 120 mV per decade can be explained by the fact that the Pt electrode surface is covered by oxides in the low overpotential region and oxide free in the high overpotential region.$^{12,19,46}$ Interestingly the transition potentials between the 60 and 120 mV per decade slopes remained virtually invariant at about +0.90 V for the four single-crystal electrodes both in the absence and presence of H$_3$PO$_4$, suggesting that the rate-determining step in ORR was not affected by the adsorption of phosphate anions.$^{47}$

Furthermore, in order to understand the effect of H$_3$PO$_4$ adsorption, it is important to deconvolute it from the concomitant poisoning effect of oxides. If we define $\Delta$ as the difference between the Tafel slopes at high ($t_h$) and low ($t_l$) current density, i.e., $\Delta = t_h - t_l$, then the difference in 0.1 M HClO$_4$ and in 0.1 M HClO$_4$ + 100 mM H$_3$PO$_4$ ($\Delta_{0}$) can be correlated to the effects of oxides formation and oxides formation plus H$_3$PO$_4$ adsorption, respectively. Therefore, the effect of H$_2$PO$_4$/C$_0$/4 adsorption ($\Phi$) may be quantified by

$$\Phi = \frac{\Delta_{0}}{\Delta_{100}}$$

From Table 1, it can be seen that $\Phi$ decreases in the order of $\Phi_{Pt(111)}$ (1.36) > $\Phi_{PtSn(111)}$ (1.17) > $\Phi_{Pt(110)}$ (1.12) > $\Phi_{Pt(100)}$ (1.07). The sequence is in line with the trend of catalytic activity drop-off from 0.1 M HClO$_4$ to the solution containing 100 mM H$_3$PO$_4$ for ORR on different Pt single crystals shown in Fig. 7.

Whereas it did not appear to affect the ORR reaction pathway ($vide$ $ante$), the adsorption of phosphate anions apparently hindered the oxide species formation and/or O$_2$ adsorption on Pt and thus impeded the ORR activity, as reflected by the observed variation of the Tafel slopes (Table 1).$^{33,44,45,48}$ Notably, the affinity of phosphate anions to the Pt surface varies with the surface crystalline structure, where stronger adsorption resulted in more significant

### Table 1 Kinetic parameters for ORR on Pt(100), Pt(110), Pt(111) and PtSn(111) single crystals electrodes in 0.1 M HClO$_4$ with the addition of different amounts of H$_3$PO$_4$

<table>
<thead>
<tr>
<th>H$_3$PO$_4$ concentration/mM</th>
<th>$n$</th>
<th>$E_\text{p}$V</th>
<th>Tafel slopes* at 900 rpm/mV per decade</th>
<th>Rate constant $k$ $(10^{-2}$ cm s$^{-1}$) at +0.7 V</th>
<th>$k$ at +0.85 V, 900 rpm/mA cm$^{-2}$</th>
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<tr>
<td>Pt(100)</td>
<td>0</td>
<td>3.96</td>
<td>0.736</td>
<td>62.9, 122.3 $(\Delta = 59.4)$</td>
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<td>1</td>
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<td>62.4, 122.0 $(\Delta = 59.6)$</td>
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<td>10</td>
<td>3.99</td>
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<td>0.38</td>
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<td>50</td>
<td>3.95</td>
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<td>60.1, 117.5 $(\Delta = 57.4)$</td>
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<td>0.783</td>
<td>67.8, 111.8 $(\Delta = 44)$</td>
<td>4.19</td>
<td>1.05</td>
</tr>
<tr>
<td>10</td>
<td>4.0</td>
<td>0.760</td>
<td>71.5, 112.7 $(\Delta = 41.2)$</td>
<td>3.14</td>
<td>0.82</td>
</tr>
<tr>
<td>50</td>
<td>3.96</td>
<td>0.754</td>
<td>73.0, 113.1 $(\Delta = 40.1)$</td>
<td>2.35</td>
<td>0.69</td>
</tr>
<tr>
<td>100</td>
<td>3.89</td>
<td>0.753</td>
<td>73.4, 113.4 $(\Delta = 40)$</td>
<td>1.78</td>
<td>0.63</td>
</tr>
</tbody>
</table>

* The Tafel slopes at low and high current density are separated by a comma, and $\Delta$ denotes the difference between these two slopes.
poisoning effects on ORR performance. In the voltammetric studies (Section 3.1), it was found that the adsorption of phosphate anions on Pt(111) was the most pronounced among the four single-crystal electrodes, probably because of the preferred adsorption of phosphate anions on the 3-fold sites as manifested in in situ X-ray absorption measurements (Fig. S1 and S2, ESI†). Whereas the adsorption on Pt(110) and Pt(100) was the weakest and yet comparable to each other, an observation consistent with the results from Tafel analyses. Similar observations were reported by Tanaka et al.19 and Kadiri et al.8

Another interesting observation is that PtSn(111), which was rich with steps and defects, appeared to exhibit enhanced resistance against phosphate poisoning as compared to Pt(111). Although the exact reason remains unclear at this point, a reasonable analogy may be made with Kuzume et al.’s study of ORR on stepped Pt surfaces,21 which showed that the $E_1$ value for ORR on Pt[$n$(111) × (111)] surfaces decreased as the step density decreased in sulfuric acid solutions.

Fig. 4 Plot of $j^{-1}$ vs. $i_l^{-1}$ for $O_2$ reduction at (a) Pt(100), (b) Pt(110), (c) Pt(111), and (d) PtSn(111) in $O_2$-saturated 0.1 M HClO$_4$. Symbols are experimental data acquired from Fig. 3 at different electrode potentials, and lines are linear regressions.

Fig. 5 RDE voltammograms of ORR at Pt(111) electrode in oxygen-saturated 0.1 M HClO$_4$ and 0.1 M HClO$_4$ + 50 mM H$_3$PO$_4$. Rotation rate 900 rpm. DC potential ramp 20 mV s$^{-1}$. 

Note that the $\text{Pt}_{n}(111)/\text{C}_2(111)$ surface is equivalent to $\text{Pt}[(n-1)(111) \times (110)]$ where (111) terraces are separated by monatomic (110) steps. With increasing concentration of the (110) steps, the adsorption of (bi)sulfate anions on the (111) terraces was hindered, because the adsorption of (bi)sulfate anions on stepped surfaces gave rise to disordered adlayers, which were less stable than the bi-dimensionally ordered adlayers formed by the adsorption of (bi)sulfate anions on (111) terraces. The fact that the geometry and properties of adsorption of $\text{H}_{3}\text{PO}_4$ and $\text{HPO}_4^{2-}$ anions on Pt surfaces are similar to those of $\text{HSO}_4$ and $\text{SO}_2^{2-}$ makes this analogy reasonable.

### 3.3 EIS study

Electrochemical impedance spectroscopy (EIS) is a sensitive and powerful technique to study the kinetics of electron-transfer processes. In recent years, EIS has been widely used in fuel cells to study the kinetics of electrode reactions on both...

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**Fig. 6** RDE voltammograms of ORR at (a) Pt(100), (b) Pt(110), (c) Pt(111), and (d) PtSn(111) electrodes in oxygen-saturated 0.1 M $\text{HClO}_4$ with the addition of various amounts of $\text{H}_3\text{PO}_4$. Current density was based on normalization of voltammetric current to the geometrical area of the electrode. Rotation rate 900 rpm. DC potential ramp 20 mV s$^{-1}$.

**Fig. 7** Kinetic current density of ORR at +0.85 V at Pt(100), Pt(110), Pt(111), and PtSn(111) single crystal electrodes in 0.1 M $\text{HClO}_4$ with and without 100 mM $\text{H}_3\text{PO}_4$. Data were obtained from Fig. 8.
Fig. 8  Mass-transport corrected Tafel plots of ORR at the (a) Pt(100), (b) Pt(110), (c) Pt(111) and (d) PtSn(111) single crystal electrodes in 0.1 M HClO₄ and 0.1 M HClO₄ + 100 mM H₃PO₄.

The anode\textsuperscript{56–58} and the cathode\textsuperscript{59–61} Here, the kinetics of the ORR on the four platinum single crystals as a function of H₃PO₄ concentration is examined by EIS. Fig. 9 shows the Nyquist complex-plane impedance spectra of ORR at +0.85 V (vs. RHE) with the four electrodes in O₂ saturated 0.1 M HClO₄ in the presence of different concentrations of H₃PO₄ (shown as figure legends). It can be seen that on all the single crystal surfaces under study, the EIS spectra exhibit a well-defined semicircle. Usually for ORR on electrodes with highly rough surfaces (such as those modified with Nafion films and metal nanostructured surfaces), there are two pronounced arcs corresponding to the charge transfer process at high frequencies and the mass transfer process at low frequencies.\textsuperscript{62–64} In the present study, the appearance of a single impedance arc suggests that the electrode process under the current experimental conditions is dominated by the interfacial kinetics of the ORR process.

It should be noted that the lower intercept of the impedance arc on the x-axis represents the total ohmic resistance of the system, and the diameter of the kinetic arc represents the charge transfer resistance of ORR on the electrode surface. It can be seen clearly that for each electrode in Fig. 9, the diameter of the arcs increases with the increase of H₃PO₄ concentration in the electrolyte. This result indicates that the electrocatalytic activity for ORR decreased by phosphate adsorption on the Pt single crystal surface. Specifically, phosphate anions acted as poisonous species by occupying the active sites on the Pt surface and hence impeded the adsorption of O₂ molecules. Furthermore, one can see that the effect of H₃PO₄ adsorption on the electrocatalytic activity for ORR is strongly dependent on the Pt single crystal surface structures. For example, in O₂-saturated 0.1 M HClO₄, the diameter of the impedance arc at Pt(111) is much smaller than that with Pt(100) or Pt(110).

Further insights on the electron-transfer dynamics were obtained by quantitative analysis of the Nyquist impedance spectra, using the equivalent circuit shown in Fig. 10 inset to fit the impedance data.\textsuperscript{61} Here, $R_\text{s}$ represents the solution (uncompensated) resistance, $C_{\text{DL}}$ and $R_{\text{CT}}$ are the double layer capacitance and charge transfer resistance, respectively.
Representative fits for the four electrodes were shown in each of the Nyquist plots in Fig. 9 (solid lines). It can be seen that all fits are excellent. From the fitting, the variation of the charge transfer resistance ($R_{CT}$) with the concentration of $\text{H}_3\text{PO}_4$ was shown in Fig. 10. It can be seen that on all electrodes $R_{CT}$ increases very quickly with increasing $\text{H}_3\text{PO}_4$ concentration and at concentrations greater than ca. 50 mM the $R_{CT}$ remains virtually invariant, suggesting that saturated adsorption of phosphate anions was reached in this concentration range.

However, the phosphate poisoning effect for ORR on the four Pt electrode surface was drastically different. In the electrolyte without $\text{H}_3\text{PO}_4$ (0 mM), $R_{CT}$ on Pt(111), PtSn(111), Pt(110), and Pt(100) is 81.3 Ω, 204.6 Ω, 292.5 Ω, and 493.7 Ω, respectively. With the introduction of $\text{H}_3\text{PO}_4$ into the electrolyte solution, $R_{CT}$ increases rather substantially. For instance, at 100 mM $\text{H}_3\text{PO}_4$, $R_{CT}$ increases to 1677.4 Ω for Pt(111), 1233.9 Ω for PtSn(111), 963.3 Ω for Pt(110), and 1545.7 Ω for Pt(100). That is, the degree of $\text{H}_3\text{PO}_4$ poisoning, reflected by the increase of $R_{CT}$, decreases in the order of $\text{Pt}(111) > \text{Pt}(100) > \text{PtSn}(111) > \text{Pt}(110)$, indicating that Pt(111) is the least active for ORR electrocatalysis in PAFC whereas Pt(110) represents the optimal crystalline surface among the four electrodes. This is consistent with the results based on voltammetric and Tafel analyses (vide ante), although the exact sequence of Pt(100) and PtSn(111) is somewhat different.
4. Conclusions

The effects of the adsorption of phosphate anions on the kinetics of ORR on Pt single crystal electrodes were examined by cyclic voltammetry and electrochemical impedance spectroscopy. The ORR kinetics was found to exhibit first-order dependence on the diffusing reactant (O2) and to proceed via a 4-electron pathway with water as the main product on all Pt single crystal electrodes under study. However, upon the introduction of H3PO4 into the electrolyte solution, the electrocatalytic performance of ORR was impeded by the adsorption of phosphate anions onto the surface active sites, which decreased in the order of Pt(111) > PtSn(111) > Pt(100) ~ Pt(110), as reflected in voltammetric and electrochemical impedance measurements. The most pronounced poisoning effect observed at Pt(111) was ascribed to the abundant 3-fold surface sites where adsorption of phosphate anions occurred, as manifested in in situ X-ray absorption spectroscopic measurements. Additionally, of note is that PtSn(111) showed enhanced tolerance towards phosphate anion adsorption by virtue of the surface steps and defects, suggesting that surface engineering might be an effective route towards the optimization of ORR electrocatalysts.

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Notes and references