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Rheology of granular materials: dynamics in a stress landscape

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We present a framework for analysing the rheology of dense driven granular materials, based on a recent proposal of a stress-based ensemble. In this ensemble, fluctuations in a granular system near jamming are controlled by a temperature-like parameter, the angoricity, which is conjugate to the stress of the system. In this paper, we develop a model for slowly driven granular materials based on the stress ensemble and the idea of a landscape in stress space. The idea of an activated process driven by the angoricity has been shown by Behringer *et al.* (Behringer *et al.* 2008 *Phys. Rev. Lett.* **101**, 268301) to describe the logarithmic strengthening of granular materials. Just as in the soft glassy rheology (SGR) picture, our model represents the evolution of a small patch of granular material (a mesoscopic region) in a stress-based trap landscape. The angoricity plays the role of the fluctuation temperature in the SGR. We determine (i) the constitutive equation, (ii) the yield stress, and (iii) the distribution of stress dissipated during granular shearing experiments, and compare these predictions with the experiments of Hartley & Behringer (Hartley & Behringer 2003 *Nature* **421**, 928–931.).

Keywords: granular; rheology; stress

1. Introduction

A striking feature of dry granular materials and other athermal systems is that they form force chain networks, in which large forces are distributed inhomogeneously into linear chain-like structures, in response to applied stress (Majmudar & Behringer 2005). A number of recent experimental studies have visualized and quantified force chain networks using carbon paper (Jaeger *et al.* 1996) and photo-elastic techniques (Howell *et al.* 1999; Veje *et al.* 1999). These studies have demonstrated that the characteristics of force chain networks are acutely sensitive to the nature of the prepared state, especially near the jamming transition (Majmudar & Behringer 2005). For example, in isotropically compressed systems, force chain networks are ramified with only short-ranged spatial correlations of the stress. In contrast, in sheared systems, aligned force chains give rise to long-ranged spatial correlations of the stress in the direction of the shear. Any theory of granular rheology therefore has to incorporate the effects of these grain-level structures on the macroscopic, collective response.

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It has long been realized that classical elasticity theory or Newtonian fluid dynamics is inadequate for describing the response of granular materials for a number of important reasons (Cates *et al.* 1998; Bouchaud 2003; Blumenfeld 2004): (i) tensile stresses are completely absent in dry granular materials, and therefore the cohesion of the granular assembly is induced by the applied stress, making the zero-stress state ill defined; (ii) there is an indeterminacy of the forces at the microscopic level due to friction and disorder; and (iii) granular materials are athermal and dissipative, so there is no established statistical framework that bridges the structure at grain scales to a continuum elasto-plastic theory at large length scales. In addition, granular solids often occur in isostatic states in which the number of degrees of freedom matches the number of constraints (Moukarzel 1998; Tkachanko & Witten 1999). It has been shown that the critical properties of the isostatic point, not elasticity theory, determine the mechanical response of these marginal solids (Wyart 2005).

A framework known as soft glassy rheology (SGR) has been used widely to understand the rheology of materials such as foams, colloids and grains (Sollich 1998). In this paper, we use a recently proposed statistical framework, the stress ensemble for describing the response of static granular media (Henkes *et al.* 2007; Henkes & Chakraborty 2009), to formulate a theory of the response of slowly (quasi-static) sheared granular media and we discuss its predictions for granular rheology.

2. Stress ensemble

As shown previously (Ball & Blumenfeld 2002; Henkes & Chakraborty 2009), the stress field of a mechanically stable granular material can be fully described by a spatially varying scalar field ψ in two dimensions and a tensorial field in higher dimensions (Henkes & Chakraborty 2009). These fields can be used to establish a rigorous conservation principle, valid for grain assemblies in mechanical equilibrium, at the grain level in two dimensions and at a continuum level in higher dimensions (Henkes & Chakraborty 2009). This conservation principle, in conjunction with a maximum entropy hypothesis (Jaynes 1957), leads to a generalization of equilibrium thermodynamics to the ensemble of mechanically stable granular states (Henkes *et al.* 2007; Henkes & Chakraborty 2009), where the new conserved tensorial quantity, $\hat{\Sigma}$ to be discussed below, plays the role of energy. The microcanonical version of this ensemble is characterized by the complexity, a measure of the number of grain configurations compatible with a given value of $\hat{\Sigma}$. The assumption of entropy maximization leads to the definition of a temperature-like intensive variable, the angoricity (Blumenfeld & Edwards 2003; Henkes & Chakraborty 2009), which is a tensor. The predictions of this stress ensemble have been remarkably successful in describing results of simulations (Henkes *et al.* 2007; Lois *et al.* submitted) and experiments in both static (Majmudar & Behringer 2005; Lois *et al.* submitted) and slowly driven granular media (Behringer *et al.* 2008). As an example, the stress ensemble framework and the constraint that all contact forces are non-negative have been used to develop a Ginzburg–Landau type action for jammed granular systems in terms of ψ . The theory has been used to predict the spatial correlations of stress in systems subjected to isotropic compression (Henkes & Chakraborty 2009)

and shear (Lois *et al.* submitted), and to construct a mean-field theory of unjamming under isotropic compression (Henkes & Chakraborty 2009). In slowly sheared granular systems, the observation of a logarithmic strengthening has been explained by the stress ensemble approach (Behringer *et al.* 2008). In this paper, we present a detailed development of the theory of slowly sheared granular matter, and make falsifiable predictions. Preliminary tests of the theory, based on comparison with experiments using photo-elastic discs (Hartley & Behringer 2003), are also presented.

(a) *Conservation principle and complexity*

To understand the mechanical response of granular materials, one needs a theoretical approach that can bridge the gap between microscopic, grain-level quantities and macroscopic, collective properties. Fluctuations are inherently related to the number of microscopic states available under a given set of macroscopic parameters. In equilibrium thermodynamics, the microcanonical entropy, or its derivatives in other ensembles, is the measure we use to calculate fluctuations and response. Conservation of energy allows for a rigorous definition of this measure since the states with different energies are not mixed by dynamics. In disordered systems such as spin glasses, the concept of complexity has been useful in formulating a framework for understanding collective properties (Bouchaud *et al.* 1996). Complexity is a measure of the number of states associated with a free-energy minimum and has proven to be a useful concept for disordered systems that have many metastable states. In mean-field models, these minima are separated by barriers that diverge in the thermodynamic limit, and one can in principle count the number of states unambiguously. Can we identify a physical variable in granular materials, which also have many metastable states (Bouchaud 2003), that is conserved by natural dynamics and therefore leads to a formalism for defining complexity? The lack of a natural dynamics in granular systems makes this a difficult proposition. In mechanically stable states, however, there is a topological conservation law that allows us to proceed to define the analogue of complexity. The topological nature implies that a change in this physical variable can be achieved only through rearrangements that involve the boundaries or the whole system. A *gedanken* experiment serves to illustrate the conservation law. If we draw an imaginary line through a grain assembly and calculate the value of the total normal force being transmitted along this line, then force balance ensures that this value remains unchanged as this imaginary line is translated perpendicular to itself across the assembly (Metzger 2008). A formal theoretical framework can be formulated using the force–moment tensor $\hat{\Sigma} = \int d^d r \hat{\sigma}(\mathbf{r})$, where $\hat{\sigma}(\mathbf{r})$ is the local stress tensor. It has been shown (Henkes *et al.* 2007; Henkes & Chakraborty 2009) that $\hat{\Sigma}$ depends only on the boundary conditions of the packing. Thus, the phase space of all mechanically stable configurations can be divided into sectors labelled by $\hat{\Sigma}$. Configurations in different sectors are disconnected under any local dynamics, and $\hat{\Sigma}$ plays a role similar to total energy in equilibrium thermodynamics or free-energy minima in spin glasses. Based on some very general principles, such as the factorization of states (Bertin *et al.* 2006), the conservation of $\hat{\Sigma}$ has been shown to lead to an intensive variable, $\hat{\alpha} = \partial S(\hat{\Sigma})/\partial \hat{\Sigma}$,

where $S(\hat{\Sigma})$ is the entropy of the sector labelled by $\hat{\Sigma}$ (Henkes & Chakraborty 2009). The inverse of $\hat{\alpha}$, the angoricity (Edwards & Blumenfeld 2007), is expected to play the same role as temperature in equilibrium thermodynamics, if processes that create granular assemblies achieve entropy maximization. Comparisons with simulations have demonstrated that a mechanically stable assembly of grains has the same value of angoricity throughout its interior, while $\hat{\sigma}(\mathbf{r})$, the local stress, fluctuates. The probability of occurrence of a microscopic state ν within a mechanically stable grain packing, maintained at an angoricity α , is

$$P_\nu = \left(\frac{1}{Z} \right) e^{-\hat{\alpha} \cdot \hat{\Sigma}_\nu}, \quad (2.1)$$

analogous to the Boltzmann distribution (Henkes & Chakraborty 2009). The state $\nu \equiv [\mathbf{r}_i, \mathbf{f}_{ij}]$, where \mathbf{r}_i denotes the positions of the grains and \mathbf{f}_{ij} the set of contact forces.

A natural question to ask is whether the similarities between temperature and angoricity extend to the dynamics of slowly deformed granular systems, situations analogous to dynamics of thermal systems close to thermal equilibrium where stochastic dynamics such as relaxational, Langevin dynamics provides a good description (Goldenfeld 1992). In the granular context, we are restricting our attention to systems that are close to granular equilibrium defined by equation (2.1). As in systems close to thermal equilibrium, we can construct a stochastic, Markov process based on the concept of detailed balance:

$$\frac{W_{\nu \rightarrow \nu'}}{W_{\nu' \rightarrow \nu}} = \frac{P_{\nu'}}{P_\nu}. \quad (2.2)$$

This is an additional assumption beyond the ones entering the construction of the stress ensemble that leads to the distribution of mechanically stable states, P_ν . Stochastic equations, based on detailed-balance conditions, lead to a Boltzmann-like distribution, equation (2.1), as the time-independent distribution of states (Goldenfeld 1992). If, however, the stress landscape (or energy landscape in a thermal ensemble) is such that complete equilibration is not possible, then the stochastic equations lead to a rate of escape over a barrier $\Delta \hat{\Sigma}$, which is the well-known Kramers rate: $e^{-\hat{\alpha} \cdot \Delta \hat{\Sigma}}$ (Kramers 1940). A simple example of a landscape where equilibration is not possible is the one that has a single local minimum and a single local maximum. In that situation, the system ‘escapes’ over the barrier with a rate prescribed by Kramers but cannot equilibrate. As will become clear from the discussion below, trap models describe systems that escape over barriers with the Kramers rate; however, subsequent sampling of traps is described by a quenched distribution and not by detailed balance. We adopt this form of activated dynamics to analyse the rheology of dense, quasi-static granular flows.

Our framework of granular rheology is, therefore, based on a rigorous conservation principle and postulates, similar to the ones adopted in the development of equilibrium thermodynamics (Callen 1960). These postulates are (i) factorizability of the dynamics-dependent frequency with which each grain configuration is accessed (Henkes & Chakraborty 2009), which allows for the definition of the angoricity, (ii) maximization of entropy, which implies equality

of the angoricity across a system in granular equilibrium and a Boltzmann-like distribution (equation (2.1)) (Henkes & Chakraborty 2009), and (iii) satisfaction of detailed balance by the rates of transition between microscopic states.

3. Trap models and rheology

As discussed earlier, the existence of a large number of different microscopic metastable states that are macroscopically equivalent puts granular materials into a wider class of previously well-studied systems, including gels, glasses, colloids, emulsions, polymers and foams. A minimal model that encapsulates these features and leads to a glass transition is the ‘trap model’ (Monthus & Bouchaud 1996). In this model, one considers a system made of independent subsystems of a certain size ξ , where each subsystem acts self-coherently and independently of the others. Their dynamics involves hopping between different metastable states aided by some kind of fluctuation. This idea of a random walk in a rugged landscape has its roots in the context of glasses (Bouchaud & Georges 1990). Below, we discuss the trap model and its generalization, which is the framework of SGR (Sollich 1998), in some detail.

(a) Bouchaud’s trap model for glasses

Monthus & Bouchaud (1996) constructed a one-element model for glasses. In this model, there exists an energy landscape of traps with various depths E . An element hops between traps when activated, where the fluctuation is assumed to be thermal. At a temperature $k_B T \equiv 1/\beta$, the probability of being in a trap with depth E at time t evolves according to

$$\frac{\partial}{\partial t} P(E, t) = -\omega_0 e^{-\beta E} P(E, t) + \omega(t) \rho(E), \quad (3.1)$$

where

$$\omega(t) = \omega_0 \langle e^{-\beta E} \rangle_{P(E,t)} = \omega_0 \int dE P(E, t) e^{-\beta E}. \quad (3.2)$$

The first term on the right-hand side (r.h.s.) of equation (3.1) is the rate of hopping out of a trap, where $e^{-\beta E}$ is the activation factor and ω_0 a frequency constant. The model assumes that choosing a new trap is independent of history, so that a new trap is chosen from a distribution of trap depths $\rho(E)$ that reflects the underlying disorder in the glassy models. Equation (3.2) gives the average hopping rate. This rate, multiplied by the distribution of trap depth, equation (3.1), gives the probability rate of choosing a new trap.

The existence of a glass transition in this model can be demonstrated as follows. Assume that the distribution function has a simple exponential tail such as

$$\rho(E) \propto e^{-\beta_0 E}, \quad (3.3)$$

where β_0 is a fixed parameter describing the disorder in the inherent energy landscape. The physical justification for having an exponentially decaying $\rho(E)$ is borrowed from systems with quenched random disorder, such as spin glasses, which use extreme value statistics (Monthus & Bouchaud 1996). (We will later

justify applying the same form of distribution to granular materials.) Then we can easily solve for the steady-state ($(\partial/\partial t)P(E, t) = 0$) solution:

$$P_{\text{eq}}(E) \propto e^{\beta E} \rho(E) \propto e^{(\beta_0 - \beta)E}. \quad (3.4)$$

Immediately, we see that equation (3.4) is non-normalizable for $\beta > \beta_0$ or $T < T_0$. This shows that, below a temperature T_0 , the system is out of equilibrium; it is non-ergodic and ages by evolving into deeper and deeper traps. Therefore, we call $T = T_0$ a point of glass transition (Monthus & Bouchaud 1996).

It should be noted that dynamical equations of the trap model do not lead to a Boltzmann distribution at T since the equations do not obey detailed balance, and specifically the traps are sampled from a quenched distribution. The escape rate from a trap is, however, determined by the Kramers process.

(b) Soft glassy rheology

In Sollich (1998), a simple phenomenological model termed ‘soft glassy rheology’ (SGR) was proposed to describe the anomalous rheological properties of ‘soft glassy materials’. The model incorporated Bouchaud’s glass trap model and an extra degree of freedom, the strain. The model assumes that a macroscopic soft glassy material can be subdivided into a large number of mesoscopic regions, each having a linear size ξ . With each mesoscopic region, one can then associate an energy E and a strain variable l , which both evolve with time. The number of mesoscopic regions must be made large enough so that ensemble averaging can be performed over them to yield macroscopic properties of soft glassy materials. Similar to Bouchaud’s trap model, the SGR model assumes that the mesoscopic regions ‘live’ in an energy landscape. This is a mean-field energy landscape in that it is not characterized by a metric and is defined only by the distribution $\rho(E)$. The new strain variable describes local elastic deformation of the mesoscopic regions, so that l contributes quadratically to the energy E . Since a strain variable was added to the SGR, the model can now describe material under imposed shear strain; also in a mean-field spirit, all mesoscopic regions respond uniformly to externally imposed shear strain. Different from Bouchaud’s trap model, thermal fluctuations are unimportant in soft materials ($k_B T$ is too small to cause structural rearrangements). In the SGR model, it is the fluctuations in the elastic energy that facilitate structural rearrangements. This fluctuation is determined by a temperature-like quantity x called the ‘noise level’. As a result, the escape rate from a trap becomes $e^{-E/x}$. The use of x is a mean-field approach to describe all interactions between mesoscopic regions.

(c) Incorporating the stress ensemble into the soft glass rheology framework

The idea of using a temperature-like quantity to replace thermal fluctuations in the SGR model is reminiscent of the stress ensemble for granular materials, where the inverse angoricity, $1/\hat{\alpha}$, plays the role of the noise, x , in SGR. The activated process in the stress ensemble is a result of a coupling between different mesoscopic regions through stress fluctuations. Each mesoscopic region can be viewed as existing in a bath of stress fluctuations, which is characterized by the angoricity, and these fluctuations can lead to activated processes analogous to those occurring in thermal systems. Since the granular material in relevant

experiments (Hartley & Behringer 2003) are sheared set-ups, and to simplify the model, we will use a scalar model that incorporates only the shear components of the stress through Γ , the deviatoric part of $\hat{\Sigma}$, and α , the shear component of the inverse-angoricity tensor (referred to as shear angoricity below), which represents the bath of stress fluctuations that the mesoscopic region is in contact with.

To adapt the SGR framework to the stress ensemble, we need to define metastability and escape processes in stress rather than to define energy space. Since the stress ensemble is based on the premise that there are many states $[\mathbf{r}_i, \mathbf{f}_{ij}]$ with a given Γ , and that these states are broadly sampled even if not with equal weights (Henkes & Chakraborty 2009), we assume that mesoscopic regions of granular assemblies can be in metastable equilibrium characterized by the Boltzmann-like distribution (equation (2.1)). A dynamics obeying detailed balance (equation (2.2)) would lead to this distribution. As mentioned earlier, the trap model and SGR, however, describe systems in the presence of quenched disorder, which define the distribution barrier heights (or equivalently trap depths) $\Delta\Gamma$ that can be crossed in a finite time. The system can escape from a metastable state represented by a trap. Stochastic equations based on the stress ensemble allow us to adopt Kramers' escape rate approach. Replacing temperature by the shear angoricity, the escape rate is therefore, $e^{-\alpha\Delta\Gamma}$, which replaces $e^{-\beta E}$ of Bouchaud's trap model. As in the original trap model framework, the exploration of the traps in stress space following this escape is assumed to be determined by the intrinsic, quenched-in distribution of trap depths, and this dynamics does not lead to the Boltzmann-like distribution, equation (2.1), in the long-time limit. The quenched-in distribution of trap depths determines whether complete equilibrium is attainable in the absence of shearing (Monthus & Bouchaud 1996). The activated process in the original trap model was thermal, and in SGR the activation is controlled by the noise x . In the scalar version of the stress ensemble, the activation is in stress space and is controlled by the angoricity α . The disorder in grain packings is represented by an intrinsic distribution of trap depths, just as in SGR or the original trap model.

We can now write a generalization of the SGR model to granular materials using the stress ensemble

$$\frac{d}{dt}P(\Gamma_m, \Gamma, t) = -\omega_0 e^{-\alpha(\Gamma_m - \Gamma)}P(\Gamma_m, \Gamma, t) + \omega(t)\rho(\Gamma_m)\delta(\Gamma). \quad (3.5)$$

Equation (3.5) describes the evolution of the probability of finding a mesoscopic region with a 'yield stress' Γ_m (again, strictly, Γ_m has the unit of force moment) and an instantaneous stress Γ . The yield stress Γ_m is used to define the depth of the trap the system is in, and since Γ is the instantaneous stress of the system, the barrier is $\Delta\Gamma = \Gamma_m - \Gamma$. The trap depth E of SGR or the trap model is therefore replaced in the stress ensemble by Γ_m .

The model accounts for the successive stress build-up and yielding events inside a mesoscopic region. When the instantaneous stress (due to external driving) is less than the yield stress, or $\Gamma < \Gamma_m$, the mesoscopic region is considered to be inside a 'trap' of depth Γ_m . It is possible to yield below the yield stress because of the activated nature of the process. At this point, the mesoscopic region 'sees' a reduced trap depth (i.e. the effective barrier height) $\Gamma_m - \Gamma$, and hence can undergo an activated process to 'hop' out of the trap with a rate proportional

to $e^{-\alpha(\Gamma_m - \Gamma)}$. Another process through which the mesoscopic regions can yield is that, if no activated process takes place while in the trap ($\Gamma < \Gamma_m$), Γ can increase due to the external applied shear up to the point $\Gamma = \Gamma_m$ where the mesoscopic region must yield, hence leaving the trap. This is the only possible process at the limit of zero angoricity ($1/\alpha \rightarrow 0$), and is not explicitly included in equation (3.5). Either way, after the mesoscopic region leaves a trap, it chooses a new trap according the distribution $\rho(\Gamma_m)$. Figure 1 shows a schematic of this dynamics of the mesoscopic region in the landscape.

The mesoscopic regions are assumed to be large enough to be treated as deforming elastically until a ‘plastic’ yield event occurs. In the trap picture, these plastic events are equated with the mesoscopic region leaving a trap. The grains are driven by a constant shear rate in the experiments (Hartley & Behringer 2003). In a mean-field approach similar to SGR, we assume that this applied shear is transmitted without decay throughout the entire material. Thus, each mesoscopic region is driven by the same shear rate. Since the mesoscopic regions deform elastically, the stress Γ increases linearly with time up to the point of yielding,

$$\sigma = \frac{1}{S}\Gamma = kl = k\dot{\gamma}t, \quad (3.6)$$

where σ is the shear stress, S is the area of the mesoscopic region and k is the elastic constant relating the shear stress to the shear strain, l . The continual process of stress build-up and yielding creates a ‘sawtooth’ pattern in time, as shown at the bottom of figure 1.

With a linear dependence of Γ on time, between yield events, we can rewrite equation (3.5) as

$$\frac{\partial}{\partial t}P(\Gamma_m, \Gamma, t) = -Sk\dot{\gamma} \frac{\partial}{\partial \Gamma}P(\Gamma_m, \Gamma, t) - \omega_0 e^{-\alpha(\Gamma_m - \Gamma)}P(\Gamma_m, \Gamma, t) + \omega(t)\rho(\Gamma_m)\delta(\Gamma). \quad (3.7)$$

The full time differential on the l.h.s. of equation (3.5) has been converted to a partial time derivative plus an advective term. The delta function in the last term on the r.h.s. of equation (3.7) means that, immediately after choosing a new trap, the mesoscopic region is in a stress-free state. See points ‘3’ and ‘5’ in figure 1. If $\dot{\gamma} = 0$, the model reduces to Bouchaud’s trap model in stress space. The steady-state distribution, $P(\Gamma_m, \Gamma) \propto e^{-\alpha(\Gamma_m - \Gamma)}\rho(\Gamma_m)\delta(\Gamma)$, is the equilibrium probability of finding a region with yield stress Γ_m . Since there is no imposed shearing, the instantaneous stress is zero.

Physically, hopping out of a trap equates the grains in the mesoscopic region making a rearrangement. These events are visible in experiments and, if analysed with photo-elastic techniques, are just the buckling or breaking of force chains. After this rearrangement, new force chains form. The newly formed force chains are characterized by a new yield stress Γ_m .

The distribution of trap depths $\rho(\Gamma_m)$ describes the disorder in the stress landscape. Theoretically, this distribution can be measured in experiments where no activated stress yielding occurs, or when stress fluctuations exist only locally, which leads to the angoricity ($1/\alpha$) being very small. This is the case when no neighbouring stress fluctuation is ‘felt’ within the mesoscopic region. This is realized experimentally (Yu 2008). In this experiment, shear is applied to a small local region that can be considered as a single mesoscopic region. The

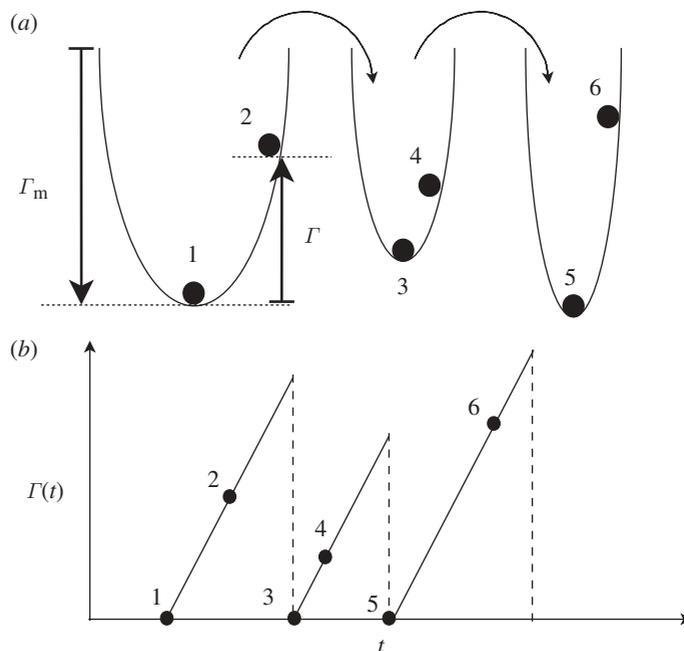


Figure 1. Schematic showing dynamics of the model in the stress landscape. The black dots with the numbering (1–6) represent the state of the mesoscopic region at different times. (a) Each ‘trap’ in the landscape is given by its depth Γ_m . The amount of stress build-up, while still in the trap, is indicated by the height Γ from the bottom of the trap. (b) For ‘1’, the mesoscopic region has just fallen into the trap and it has no stress build-up, so it has value of $\Gamma = 0$. As constant shear stress is applied to the mesoscopic region, its stress starts to build up. In (a) ‘2’ is at a height from the bottom of the trap; and in (b) ‘2’ has increased linearly in value from ‘1’. Next, via the activated process, the mesoscopic region makes a jump from one trap to another. Note that the traps shown are intentionally made disjoint, to indicate that every trap can be accessed from every other trap and there is no specific connectivity.

stress yielding is due entirely to the driving. The distribution of yield stresses in this case is exactly $\rho(\Gamma_m)$. Generally, in sheared granular systems (e.g. Geng & Behringer 2005), we can get an idea of the form of $\rho(\Gamma_m)$ by looking at the tail of the stress distribution (large Γ). This is because large Γ values are accessed only when an activated process has not occurred to make the system yield. They are also states that sample large Γ_m values.

Activated dynamics in stress space is not an entirely new concept. It was proposed by Eyring (Glasstone *et al.* 1941), and has been applied to analyse the velocity profile in dense granular flows (Pouliquen & Gutfraind 1996). In our work, the stress activation is a natural consequence of the ensemble based on angoricity.

We can write down a general solution to equation (3.7) regardless of the distribution of trap depth $\rho(\Gamma_m)$. Equation (3.7) can be simplified with a change of variable $\delta\Gamma = \Gamma(t) - Sk\dot{\gamma}t$, which does not explicitly depend on time (away from the points of yielding). This eliminates the advective term in equation (3.7),

which then becomes

$$\frac{\partial}{\partial t} P(\Gamma_m, \delta\Gamma, t) = -\omega_0 e^{-\alpha[\Gamma_m - (\delta\Gamma + Sk\dot{\gamma}t)]} P(\Gamma_m, \delta\Gamma, t) + \omega(t)\rho(\Gamma_m)\delta(\delta\Gamma + Sk\dot{\gamma}t). \quad (3.8)$$

A better intuition can be gained by solving equation (3.8), while ignoring the second term on its r.h.s., and the solution is

$$P(\Gamma_m, \delta\Gamma, t) = P_0(\Gamma_m, \delta\Gamma) \exp\left(-\frac{\omega_0}{e^{\alpha\Gamma_m}} \int_0^t dt' e^{\alpha(\delta\Gamma + Sk\dot{\gamma}t')}\right). \quad (3.9)$$

The solution is just an exponential decay with the time interval replaced by a time integral. This is the ‘effective time interval’ defined in Sollich (1998) as

$$Z(t, t'; \delta\Gamma) \equiv \int_{t'}^t dt'' e^{\alpha(\delta\Gamma + Sk\dot{\gamma}t'')}. \quad (3.10)$$

In essence, the behaviour of an element in a trap is just that of a decaying process, approximately $e^{-t/\tau}$. For us, the time is replaced by the effective time interval $Z(t, 0; \delta\Gamma)$, which grows much faster than linearly, and the mean lifetime is $\tau = \omega_0^{-1} e^{\alpha\Gamma_m}$. After the stress collapse of the first build-up occurring at t' , the element escapes into another trap chosen from the distribution $\rho(\Gamma_m)$ and starts to undergo its own decay process via the effective time. Therefore, the full solution is

$$P(\Gamma_m, \delta\Gamma, t) = P_0(\Gamma_m, \delta\Gamma) \exp\left(-\frac{\omega_0}{e^{\alpha\Gamma_m}} Z(t, 0; \delta\Gamma)\right) + \int_0^t dt' \omega(t')\rho(\Gamma_m)\delta(\delta\Gamma + Sk\dot{\gamma}t') \exp\left(-\frac{\omega_0}{e^{\alpha\Gamma_m}} Z(t, t'; \delta\Gamma)\right). \quad (3.11)$$

4. Theoretical predictions and comparison with experiments

Slow shearing experiments correspond to the long-time or steady-state limit of our model. In such a limit, we find a solution that is time independent,

$$P(\Gamma_m, \Gamma) = \rho(\Gamma_m) \exp\left(-\frac{\omega_0}{Sk\dot{\gamma}\alpha e^{\alpha\Gamma_m}} (e^{\alpha\Gamma} - 1)\right), \quad (4.1)$$

subject to normalization. We now have an ensemble distribution function of Γ with which we can calculate the constitutive relation by taking the ensemble average of Γ . Using equation (3.6) we can write

$$\sigma = \frac{1}{S} \langle \Gamma \rangle_{P(\Gamma_m, \Gamma)} = \frac{1}{S} \int_0^\infty d\Gamma \Gamma \int_0^\infty d\Gamma_m P(\Gamma_m, \Gamma). \quad (4.2)$$

We first consider the simplest case of a distribution of trap depths, $\rho(\Gamma_m) = \delta(\Gamma_m - \Gamma_0)$ with $\Gamma_0 > 0$. Although not a physical description, this choice will prove to be a fruitful exercise. Equation (4.2) becomes

$$\sigma = \frac{1}{\alpha S} \frac{\int_1^\infty dy y^{-1} \log(y) e^{-Ay}}{E_1(A)}, \quad (4.3)$$

where the constant $A = (\omega_0/Sk\dot{\gamma}\alpha) e^{-\alpha\Gamma_0}$ and E_1 is the exponential integral with $n = 1$. In the interesting limit of $A \rightarrow 0$ or $\dot{\gamma} \gg \omega_0 e^{-\alpha\Gamma_0}/Sk\alpha$, we can expand equation (4.3) to $\mathcal{O}(A^0)$ and obtain

$$\sigma = \frac{1}{2\alpha S} \log(\dot{\gamma}) + \frac{1}{2\alpha S} \left(\log\left(\frac{Sk\alpha}{\omega_0}\right) + \alpha\Gamma_0 - 2\gamma_c \right), \quad (4.4)$$

where $\gamma_c \simeq 0.577$ is the Euler–Mascheroni constant. The leading behaviour of the constitutive equation has a logarithmic dependence on the shear rate. This agrees qualitatively with experiments (Hartley & Behringer 2003; Behringer *et al.* 2008). In the opposite limit $A \rightarrow \infty$ or $\dot{\gamma} \ll \omega_0 e^{-\alpha\Gamma_0}/Sk\alpha$, we can expand equation (4.3) to $\mathcal{O}(1/A)$ and obtain

$$\sigma = \frac{k e^{\alpha\Gamma_0}}{\omega_0} \dot{\gamma}. \quad (4.5)$$

This reveals a Newtonian regime with viscosity $\eta = k e^{\alpha\Gamma_0}/\omega_0$.

(a) *Constitutive equation for the exponentially decaying distribution of trap depths*

As discussed in the previous section, an exponentially decaying $\rho(\Gamma_m)$ is a better description of granular experiments as well as spin glasses. We use

$$\rho(\Gamma_m) = \frac{1}{\alpha_0} e^{-\alpha_0\Gamma_m}, \quad (4.6)$$

where α_0 is a constant measuring the disorder in the stress landscape. Inserting equation (4.6) into equation (4.2), we obtain

$$\sigma = \frac{1}{\alpha S} \log\left(\frac{Sk\dot{\gamma}\alpha}{\omega_0}\right) + \frac{1}{\alpha S} \frac{\int_0^\infty dW (1/W^x) [\log(W + \omega_0/Sk\dot{\gamma}\alpha)/(W + \omega_0/Sk\dot{\gamma}\alpha)] \gamma(x, W)}{\int_0^\infty dW [1/W^x (W + \omega_0/Sk\dot{\gamma}\alpha)] \gamma(x, W)}, \quad (4.7)$$

where x is given by the ratio $x = \alpha_0/\alpha$, and $\gamma(x, W)$ is the lower incomplete gamma function. Again we are interested in the limit of $\dot{\gamma} \gg \omega_0/k\alpha$ and the constitutive equation is approximated to the zeroth order as

$$\sigma = \frac{2}{\alpha S} \log(\dot{\gamma}) + \frac{1}{\alpha S} \log\left(\frac{Sk\alpha^2}{\omega_0\alpha_0}\right). \quad (4.8)$$

This shows that the logarithmic dependence of shear stress on shear rate is independent of the form of the distribution $\rho(\Gamma_m)$ used. In figure 2, we compare this result with the data obtained in two-dimensional Couette granular shearing experiments (Hartley & Behringer 2003).

In the opposite limit, $\dot{\gamma} \ll \omega_0/Sk\alpha$, we again obtain a Newtonian regime

$$\sigma = \frac{k}{\omega_0} \frac{x-1}{x-2} \dot{\gamma}. \quad (4.9)$$

The viscosity diverges at $x = 2$; the same result is obtained in SGR (Sollich 1998).

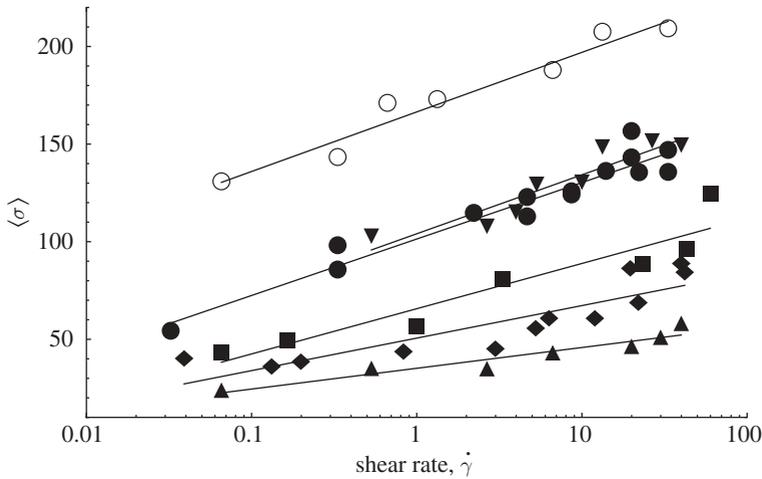


Figure 2. Linear-log plot of mean stress in a segment of two-dimensional granular Couette experiment containing roughly 200 particles versus shear rate. Data, taken from Hartley & Behringer (2003), are given for different packing fractions, which are given relative to the critical packing fraction ϕ_c where the system loses all rigidity. Various symbols are: upward triangles, $\phi = \phi_c + 0.00035$; diamonds, $\phi = \phi_c + 0.00141$; squares, $\phi = \phi_c + 0.00528$; solid circles, $\phi = \phi_c + 0.00528$; downward triangles, $\phi = \phi_c + 0.00800$; and open circles, $\phi = \phi_c + 0.01373$.

(b) Yield stress

When $\dot{\gamma} = 0$, the stress has a finite value for $\alpha_0 < \alpha$ or $x < 1$. This is the yield stress, and it vanishes at the same point as the glassy transition point in Bouchaud's trap model. The yield stress can be calculated by inserting $\dot{\gamma} = 0$ into equation (4.2). Performing the Γ integral first in equation (4.2), we obtain

$$\sigma = \frac{1}{\alpha S} \frac{\int_0^{\omega_0/Sk\dot{\gamma}\alpha} dz z^{x-1} G(z)}{\int_0^{\omega_0/Sk\dot{\gamma}\alpha} dz z^{x-1} E_1(z)}, \quad (4.10)$$

where $G(z)$ is the Meijer G -function:

$$G\left(z \left| \begin{matrix} 1, 1 \\ 0, 0, 0 \end{matrix} \right. \right) = \int_1^\infty dy \frac{\log(y)}{y} \exp(-z(y-1)).$$

When $\dot{\gamma} = 0$, equation (4.10) can be calculated exactly:

$$\sigma_y \equiv \sigma(\dot{\gamma} = 0) = -kx\gamma_c - kx \frac{d}{dx} \log(\Gamma(x)). \quad (4.11)$$

A plot of the yield stress is shown in figure 3.

(c) Distribution of stress drops

It is often easier to measure the distribution of stress drops in experiments than the stresses themselves, and the distribution provides a more stringent test of theoretical frameworks. In a trap with depth Γ_m , the probability to build up

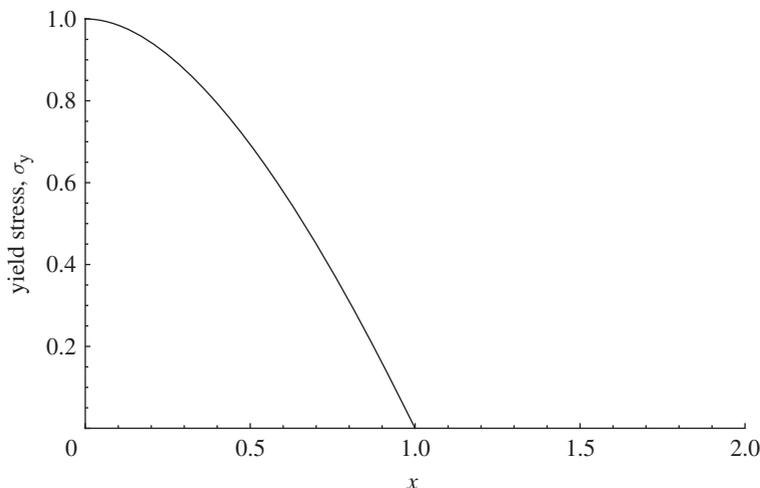


Figure 3. Theoretical plot of the yield stress, equation (4.11), as a function of $x = \alpha_0/\alpha$. It vanishes at $x = 1$.

the stress to a value of Γ is given by (cf. equation (4.1))

$$P_s(\Gamma) \propto \exp\left(-\frac{\omega_0}{Sk\dot{\gamma}\alpha} e^{\alpha\Gamma_m} (e^{\alpha\Gamma} - 1)\right).$$

On the other hand, as the stress grows, it is increasingly likely to fail due to the activated process. The rate of failure is given by

$$P_f(\Gamma) \propto e^{-\alpha(\Gamma_m - \Gamma)}.$$

Then, the probability of building the stress up to Γ and then to fail at this point is given by the product $P_s \times P_f$. Since the stress falls to zero after a failure event, we can call the magnitude of the stress drop $\Delta\Gamma$. Its distribution for a given $\rho(\Gamma_m)$ is proportional to

$$P(\Delta\Gamma) \propto P_s(\Delta\Gamma)P_f(\Delta\Gamma).$$

Also we need to sum over all possible traps:

$$P(\Delta\Gamma) \propto \int d\Gamma_m \rho(\Gamma_m) P_s(\Delta\Gamma) P_f(\Delta\Gamma). \quad (4.12)$$

Using the exponentially decaying distribution $\rho(\Gamma_m) = e^{-\alpha_0\Gamma_m}$, we obtain the normalized distribution

$$P(\Delta\Gamma) = x\alpha \left(\frac{\omega_0}{Sk\dot{\gamma}\alpha}\right)^{-x} e^{\alpha\Delta\Gamma} (e^{\alpha\Delta\Gamma} - 1)^{-1-x} \gamma\left(1 + x, \frac{\omega_0}{Sk\dot{\gamma}\alpha} (e^{\alpha\Delta\Gamma} - 1)\right), \quad (4.13)$$

where we see the lower incomplete gamma function again. There are three regimes in which equation (4.13) can be simplified. First, using the limit $y^{-\beta}\gamma(\beta, y) \rightarrow 1/\beta$ as $y \rightarrow 0$, we obtain

$$P(\Delta\Gamma) = \left(\frac{x}{x+1}\right) \frac{\omega_0}{Sk\dot{\gamma}} e^{\alpha\Delta\Gamma} \quad \text{when} \quad \frac{\omega_0}{Sk\dot{\gamma}\alpha} (e^{\alpha\Delta\Gamma} - 1) \rightarrow 0. \quad (4.14)$$

Table 1. Results of fitting equation (4.15) to the experimental data of Hartley & Behringer (2003).

$\phi - \phi_c$	strain rate $\dot{\gamma}$ (mHz)	α	α_0	$x = \alpha_0/\alpha$
0.0053	0.3318	0.7	0.32473	0.4639
	2.2148	1.0	0.2339	0.2339
	4.6568	1.4	0.19054	0.1361
	8.6491	2.0	0.1624	0.0812
	13.9722	4.0	0.1244	0.0311
0.0014	0.0390	0.6	0.34872	0.5812
	0.1322	0.6	0.3279	0.5465
	0.1987	0.6	0.34566	0.5761
	0.8308	0.7	0.3115	0.4450
	2.9933	0.9	0.25857	0.2873
	5.2490	1.0	0.2366	0.2366
	6.3203	1.2	0.21372	0.1781
	11.9760	1.3	0.18083	0.1391
	19.6279	4.0	0.1428	0.0357
	21.9567	6.0	0.1836	0.0306
	0.0657	0.5	0.30135	0.6027
	0.5314	0.5	0.26265	0.5253
	2.6607	0.7	0.24444	0.3492
	6.6529	0.8	0.1928	0.2410
	11.9760	1.1	0.17061	0.1551
19.9606	1.2	0.17316	0.1443	
29.9413	1.2	0.15372	0.1281	
39.9221	1.5	0.1344	0.0896	
0.0088	0.5314	0.5	0.2375	0.4750
	2.6607	0.9	0.25146	0.2794
	3.9848	1.0	0.2282	0.2282
	5.3155	1.2	0.21312	0.1776
	9.9998	1.6	0.1776	0.1110
	13.3068	2.5	0.163	0.0652
0.0137	0.0657	0.4	0.25472	0.6368
	0.3318	0.5	0.33945	0.6789
	0.6645	0.5	0.27425	0.5485
	1.3299	0.5	0.26505	0.5301
	2.6607	1.3	0.33384	0.2568
	3.3260	1.4	0.24458	0.1747
	6.6529	1.5	0.2097	0.1398
	13.3068	4.0	0.1776	0.0444

The opposite limit for the lower incomplete gamma function is $\gamma(\beta, y) \rightarrow (\beta - 1)!$ as $y \rightarrow \infty$, for which we obtain

$$P(\Delta\Gamma) = \alpha x^2 (x - 1)! \left(\frac{\omega_0}{Sk\dot{\gamma}\alpha} \right)^{-x} e^{\alpha\Delta\Gamma} (e^{\alpha\Delta\Gamma} - 1)^{-1-x}$$

when $\frac{\omega_0}{Sk\dot{\gamma}\alpha} (e^{\alpha\Delta\Gamma} - 1) \rightarrow \infty$. (4.15)

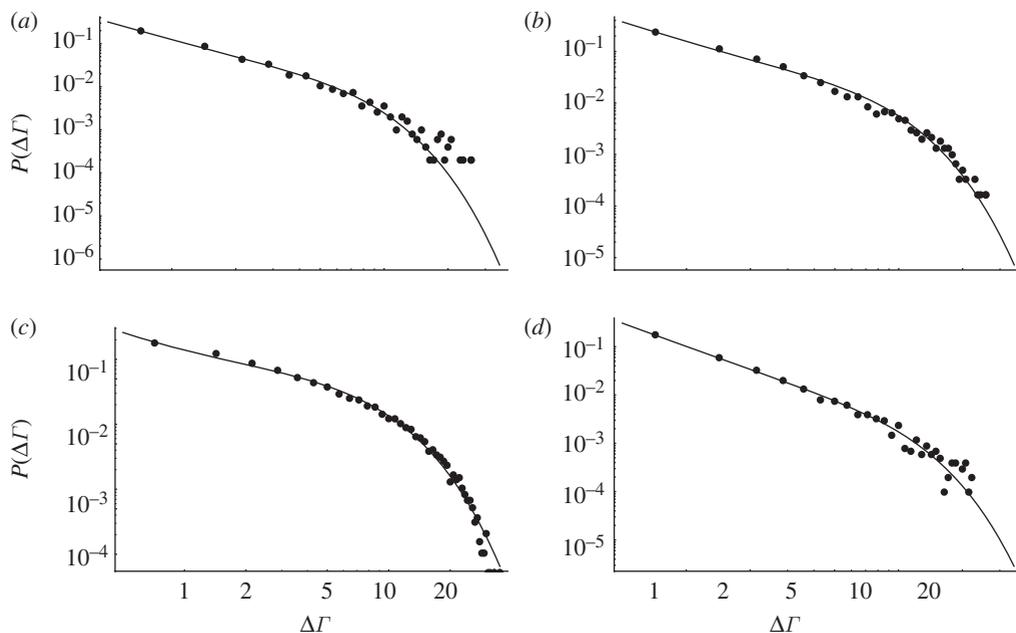


Figure 4. Log-log plot of experimental distribution of stress drops and its fitting to equation (4.15). The experimental and fitting parameters are: (a) $\dot{\gamma} = 0.3318$ mHz, $\phi - \phi_c = 0.0053$, $\alpha = 0.70$, $\alpha_0 = 0.3247$; (b) $\dot{\gamma} = 0.9972$ mHz, $\phi - \phi_c = 0.0053$, $\alpha = 0.7$, $\alpha_0 = 0.2660$; (c) $\dot{\gamma} = 5.3155$ mHz, $\phi - \phi_c = 0.0088$, $\alpha = 1.2$, $\alpha_0 = 0.2131$; and (d) $\dot{\gamma} = 0.0657$ mHz, $\phi - \phi_c = 0.0137$, $\alpha = 0.4$, $\alpha_0 = 0.2547$.

Finally, the more extreme case of equation (4.15) as $\Delta\Gamma \rightarrow \infty$ is

$$P(\Delta\Gamma) = \alpha x x! \left(\frac{\omega_0}{Sk\dot{\gamma}\alpha} \right)^{-x} e^{-\alpha_0\Delta\Gamma} \quad \text{when } \Delta\Gamma \rightarrow \infty. \quad (4.16)$$

In equation (4.16), we recover the exponential distribution of depths.

With an analytical form of $P(\Delta\Gamma)$, we can compare and fit to data obtained by Hartley & Behringer (2003). The data are fitted to the form in equation (4.15) and therefore contain two fitting parameters: α and α_0 . Table 1 shows the fitting results. In figure 4, four fitting results are plotted and compared with the data. As seen from table 1, both α and α_0 are functions of the shear rate and the packing fraction. The data fitting gives us insights into the relationship between the shear anisotropy ($x = \alpha_0/\alpha$), the shear rate ($\dot{\gamma}$) and the packing fraction $\phi - \phi_c$. We can see this relation in a plot of x versus $\dot{\gamma}$ in figure 5 for different $\phi - \phi_c$. The first observation one makes is that x is controlled by both $\dot{\gamma}$ and $\phi - \phi_c$, implying that the SGR assumption of x being independent of the shear rate does not apply to these measurements in sheared, granular media. In addition, over a decade of $\phi - \phi_c$ values and three decades of $\dot{\gamma}$, x seems to be described by a scaling form:

$$x = f_+ \left(\frac{\dot{\gamma}}{|\phi - \phi_c|^\Delta} \right), \quad (4.17)$$

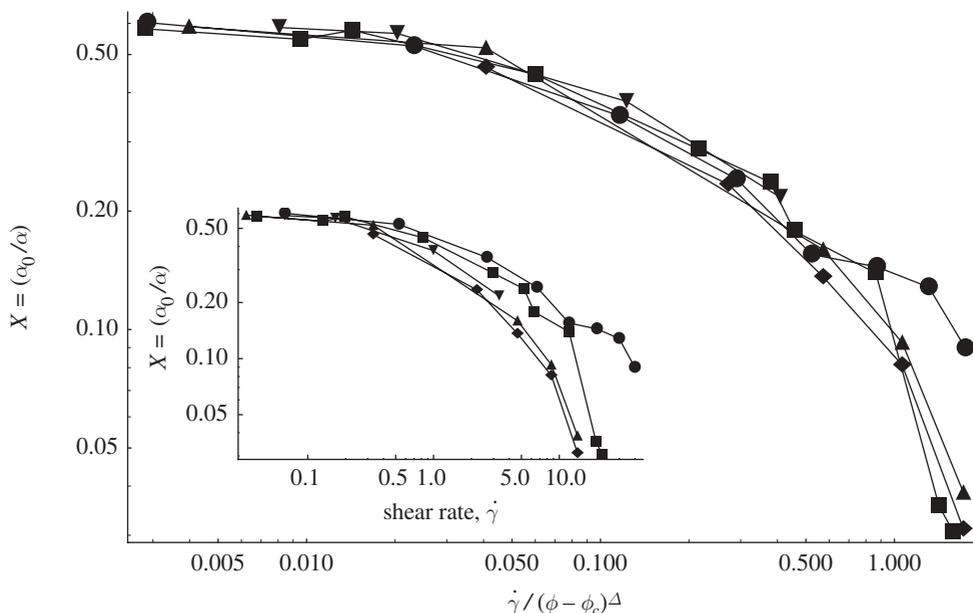


Figure 5. Log-log plot of the scaled relationship (equation (4.17)). The scaling exponent is found to be: $\Delta = -0.4$. The data points are: circles, $\phi - \phi_c = 0.0004$; squares, $\phi - \phi_c = 0.0014$; diamonds, $\phi - \phi_c = 0.0053$; upward triangles, $\phi - \phi_c = 0.0053$; and downward triangles, $\phi - \phi_c = 0.0053$. Inset: Unscaled log-log plot of x versus $\dot{\gamma}$.

with $\Delta \simeq -0.4$, and the scaling function (for $\phi \geq \phi_c$) $f_+(z)$ being a decreasing function of its argument, z . The dependence on $\dot{\gamma}$ is, however, weaker than the dependence on $\phi - \phi_c$. Although better statistics are needed to pin down the scaling form, it is intriguing to examine its consequences as such. The scaling itself hints at $(\phi - \phi_c = 0, \dot{\gamma} = 0)$ being a critical point, which is consistent with the observations of Olsson & Teitel (2007), and the idea of point J being a special point in the jamming phase diagram (Liu & Nagel 1998). At zero shear rate, x becomes independent of $\phi - \phi_c$, and the scaling form is, therefore, consistent with the packings possessing a yield stress for all $\phi > \phi_c$. From the perspective of x being the fluctuation temperature, the observation is that stress fluctuations grow as ϕ approaches ϕ_c from above. Experiments do indicate growing fluctuations as a system approaches the unjamming transition below which it cannot sustain shear (Howell *et al.* 1999).

5. Conclusion

A stress-based statistical ensemble has been used in conjunction with the idea of a stress landscape to construct a model for granular rheology. Comparisons of the model with experiments on slowly sheared granular media indicate that the theoretical framework provides a semiquantitative description of the stress response. The framework is an extension of the SGR approach, and the stress ensemble provides a natural definition for the fluctuation temperature.

This fluctuation temperature can be obtained by fitting the theoretical predictions to experiments, and the results show that fluctuations are largest at the special point, point J, where the shear rate goes to zero and the packing fraction approaches the critical value below which one cannot construct a mechanically stable state of grains. In future work, we intend to extend the model to include volume fluctuations, since Reynolds' dilatancy is one of the signature properties of granular matter (Reynolds 1885). A fundamental premise of the stress ensemble framework is that, just as in thermal equilibrium the temperature is equalized, for a system in granular equilibrium, the angoricity is the same everywhere inside the system. Comparison with simulations has shown this to be true for a system under pure compression (Henkes *et al.* 2007). If we assume the same to be true for the shear angoricity x , then, according to equation (4.17), the local packing fraction has to adjust to the local shear rate and regions of increased shear rate have lower packing fractions. An interesting question is to ask whether this is the origin of Reynolds' dilatancy.

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